



VERIFICATION OF TRANSLATION

I, named below, do hereby declare that I am knowledgeable in the English language and in Japanese language, and that I believe the attached document in the English language is a full and faithful translation of the specification in Japanese language filed in the Japanese Patent Office.

I hereby declare that all statements made herein of my own knowledge are true and that all statements made on information and belief are believed to be true; and further that these statements were made with the knowledge that willful false statements and the like so made are punishable by fine or imprisonment, or both, and that such willful false statements may jeopardize the validity of the application or any patent issued thereon.

Japanese Application No: 2002-086649

Filing date of the Japanese Application: March 26, 2002

Language of original filing Japanese

Signature of verifier:

Yoshiyuki Nishi

Name of verifier:

Yoshiyuki Nishi

Post office address of verifier:

1-17-1-704 Sugita, Isogo-ku, Yokohama-shi

Kanagawa-ken 235-0033 Japan

Date of signature:

July 10, 2007

DETAILED DESCRIPTION OF THE INVENTION

[0001]

TECHNICAL FIELD

5 The present invention relates to a luminescent material for scintillators which comprises a single crystal of an Yb-containing mixed-crystal oxide having a base crystal consisting of a garnet single crystal or a borate single crystal.

[0002]

10 BACKGROUND ART

Heretofore, a typical luminescent material for use as scintillators, as represented by Ce : GSO (Gd_2SiO_5), Ce : LSO (Lu_2SiO_5) or Ce : YAP (YAlO_3), has been based on the $5d \rightarrow 4f$ transition of Ce.

Further, patent publications disclose a Cr-activated Gd-Ga, Gd-Sc-Ga or Gd-Sc-Al garnet, a
15 Ce or Nd-activated Y-Al garnet (Japanese Patent Laid-Open Publication No. 04-289483), and a Tb-containing garnet (Japanese Patent Laid-Open Publication Nos. 07-149599 and 10-1396).

[0003]

PROBLEM(S) TO BE SOLVED BY THE INVENTION

20 The conventional luminescent materials for scintillators have the following characteristics as primary problems to be solved.

[Concentration Quenching] While Ce: GSO or Ce: LSO, which is used as a luminescent material for scintillators, has higher emission intensity as the content of Ce serving as a luminescent element is increased, the luminescent material based on the $5d \rightarrow 4f$ transition of Ce
25 exhibits strong concentration quenching when the content of Ce becomes greater than 1 atomic%, and loses any scintillator effect.

[Segregation] Ce has a large ionic radius next to La among rare-earth ions, and this

excessively large ionic radius as compared to those of typical rare-earth ions (Y, Gd, Lu) in a base crystal causes a very low segregation coefficient of Ce. Consequently, the concentration of Ce is varied in a single-crystal growth direction. The resulting variation in properties becomes a major issue in application to high-accuracy PET (Positron Emission Tomography) or the like.

5 [Emission Wavelength] The emission generated by the luminescent material has a short wavelength of 370 nm, and thereby it is required to use a photomultiplier tube as a photodetector. This leads to a technical limit concerning inapplicability of any semiconductor photodiode having a resolution 30 to 40 times greater than that of the photomultiplier tube.

10 [0004]

MEANS FOR SOLVING THE PROBLEM

The inventors found that a garnet or borate single crystal containing Tb as a mixed-crystal component is effective as a high-accuracy scintillator, because it can generate scintillation emission based on the transition to a charge transfer state (CTS) between Yb (Ytterbium) and O (oxygen) 15 while maintaining low concentration quenching at an Yb concentration of 40 atomic% or less to achieve enhanced brightness, and has a short luminescent-decay time at room temperature, and an emission wavelength falling within the sensitivity of the semiconductor photodiode.

[0005] Specifically, the present invention provides a luminescent material for scintillators, comprising a single crystal of an Yb-containing mixed-crystal oxide which has a composition 20 represented by either one of $R_3Al_5O_{12}$, $R_3Ga_5O_{12}$ and $Li_6R(BO_3)_3$, wherein R is a mixture of Yb and either one of Y, Gd and Lu, and said Yb serves as an element capable of forming an optically active state called CTS together with a neighboring negative ion (oxygen ion).

The present invention further provides a luminescent material for scintillators, comprising a single crystal of an Yb-containing mixed-crystal oxide which has a composition represented by 25 either one of $La_3R_2Ga_3O_{12}$ and $Gd_3R_2Ga_3O_{12}$, wherein R is a mixture of Yb and either one of Y, Gd and Lu, and said Yb serves as an element capable of forming an optically active state called CTS together with a neighboring negative ion (oxygen ion).

In the above luminescent material of the present invention, the molar ratio of either one of Y, Gd and Lu to Yb in the R may satisfy the conditions expressed by the following formulas:

$$1.04x + 1.02y \leq 1.03; x + y = 1; 0 < x < 1; \text{ and } 0 < y < 1$$

, wherein x is a molar ratio of Yb, and y is a molar ratio of either one of Y, Gd and Lu.

5 [0006] More specifically, the above luminescent material includes:

an aluminum garnet single crystal selected from the compound group consisting of (Y, Yb)₃Al₅O₁₂, (Gd, Yb)₃Al₅O₁₂ and (Lu, Yb)₃Al₅O₁₂;

a gallium garnet single crystal selected from the compound group consisting of (Y, Yb)₃Ga₅O₁₂, (Gd, Yb)₃Ga₅O₁₂ and (Lu, Yb)₃Ga₅O₁₂;

10 a borate single crystal selected from the compound group consisting of Li₆(Y, Yb)(BO₃)₃, Li₆(Gd, Yb)(BO₃)₃ and Li₆(Lu, Yb)(BO₃)₃;

[0007] a gallium garnet single crystal selected from the compound group consisting of La₃(Y, Yb)₂Ga₃O₁₂, La₃(Gd, Yb)₂Ga₃O₁₂ and La₃(Lu, Yb)₂Ga₃O₁₂; and

15 a gallium garnet single crystal selected from the compound group consisting of Gd₃(Y, Yb)₂Ga₃O₁₂, Gd₃(Gd, Yb)₂Ga₃O₁₂ and Gd₃(Lu, Yb)₂Ga₃O₁₂.

[0008]

BEST MODE FOR CARRYING OUT THE INVENTION

A luminescent material of the present invention selected from the compound group consisting

20 of (Y, Yb)₃Al₅O₁₂, (Gd, Yb)₃Al₅O₁₂, (Lu, Yb)₃Al₅O₁₂, (Y, Yb)₃Ga₅O₁₂, (Gd, Yb)₃Ga₅O₁₂, (Lu, Yb)₃Ga₅O₁₂, La₃(Y, Yb)₂Ga₃O₁₂, La₃(Gd, Yb)₂Ga₃O₁₂, La₃(Lu, Yb)₂Ga₃O₁₂, Gd₃(Y, Yb)₂Ga₃O₁₂, Gd₃(Gd, Yb)₂Ga₃O₁₂, Gd₃(Lu, Yb)₂Ga₃O₁₂, Li₆(Y, Yb)(BO₃)₃, Li₆(Gd, Yb)(BO₃)₃ and Li₆(Lu, Yb)(BO₃)₃ has a base crystal consisting of a garnet or borate single crystal formed of an oxide of Al, Ga or B which is a group-XIII element.

25 [0009] These compounds contain Yb serving as an element capable of forming an optically active state called CTS together with a neighboring negative ion (oxygen ion).

Ce, Nd, Pr or Cr, which is an additive of the luminescent materials for scintillators disclosed

in the patent publications described in the above "Background Art", or Yb, which is an additive for use in laser emission, forms an excited state inside the nucleus thereof.

By contrast, in the luminescent material of the present invention, the excited state corresponds to the CTS (charge transfer state) which is not formed inside the nucleus of Yb but produced by the 5 interaction between Yb and O (oxygen), and the emission is generated through the CTS.

[0010] FIG. 1 is a schematic diagram showing the coordination of oxygen to positive ions, in a compound represented by the following composition formula: $\{C_3\}[a_2](d_3)O_{12}$. In case of $R_3Al_5O_{12}$ or $R_3Ga_5O_{12}$, Yb is located in the dodecahedral site. In case of $La_3R_2Ga_3O_{12}$ or $Gd_3R_2Ga_3O_{12}$, Yb is located in the octahedral site, and La and Gd are located in the dodecahedral 10 site.

[0011] Generally, the rare earth site can contain more than just one ion or up to about 1.03 ions. Thus, the molar ratio of either one of Y, Gd and Lu to Yb in the above R is set to satisfy the conditions expressed by the following formulas: $1.04x + 1.02y \leq 1.03$; $x + y = 1$; $0 < x < 1$; and $0 < 15 y < 1$, wherein x is a molar ratio of Yb, and y is a molar ratio of either one of Y, Gd and Lu. The highest luminous efficiency can be obtained when $0.10 < x < 0.20$.

[0012] The luminescent material of the present invention can be prepared by the following single crystal growth process.

[$R_3Al_5O_{12}$ Garnet]

Yttrium oxide (Y_2O_3), gadolinium oxide (Gd_2O_3), ytterbium oxide (Yb_2O_3), lutetium oxide 20 (Lu_2O_3) and aluminum oxide (a- Al_2O_3) which have a purity of 5N (99.999 %) are used as starting materials. These starting materials are checkweighed and mixed to obtain an intended composition, and then molded and sintered to form a raw material of a luminescent material.

[0013] A single crystal is grown using a modified micro pulling-down apparatus based on high-frequency induction heating. The micro pulling-down apparatus is an unidirectional 25 solidification/crystal growth apparatus comprising a crucible furnace, a seed holder for holding a seed crystal to be brought into contact with a molten liquid flowing out from a small hole formed in the bottom of the crucible furnace, a shift mechanism for moving the seed holder downwardly, a

control unit for controlling the moving speed of the shift mechanism, and an induction-heating device for heating the crucible furnace.

[0014] The crucible furnace is made of iridium metal or iridium alloy, and an afterheater, which is a heat generator made of iridium metal or iridium alloy, is disposed around the outer periphery of the

5 bottom of the crucible furnace. The induction-heating device is operable to adjust the heating value of the crucible furnace and the afterheater so as to control the heating temperature of the solid-liquid boundary phase of the molten liquid to be extracted from the small hole in the bottom of the crucible furnace.

[0015] In this apparatus, the small hole is provided in a plural number. The small holes are

10 designed to have a diameter ($400 \mu\text{m}\varphi$ or less, preferably in the range of 200 to 300 $\mu\text{m}\varphi$, for oxide eutectic materials) which allows an effective segregation coefficient K_{eff} to be about 1 and prevents the molten liquid from dripping off therethrough, and arranged to allow the molten liquids falling from the small holes to converge or interflow before they are brought into contact with the seed crystal.

15 [0016] With the use of the above apparatus, the above molded/sintered raw material is put in the crucible furnace, and the air in the crucible furnace is evacuated to control the atmosphere in the furnace. Then, high-purity Ar gas (99.99 %) is introduced in the furnace to form an inert gas atmosphere in the furnace, and a high-frequency power is gradually applied to a high-frequency induction-heating coil to heat the crucible furnace so as to completely melt the raw material in the

20 crucible furnace. Preferably, the high-frequency power is continuously applied for about 2 hours to obtain homogeneity in composition of the molten liquid.

[0017] Subsequently, a crystal is grown through the following steps.

The seed crystal [YAG (Yttrium Aluminum Garnet) single crystal cleaved along the <111> direction] is gradually lifted at a given speed to bring the tip of the seed crystal into contact with the

25 small holes formed in the bottom of the crucible furnace (and arranged to allow the molten liquids falling from the small holes to converge or interflow before they are brought into contact with the seed crystal). After the tip of the seed crystal sufficiently fits in with the small holes, a

pulling-down shaft is moved downward while adjusting the temperature of the molten liquid to grow the crystal. When the entire molten liquid is consumed, and the prepared raw material is fully crystallized, the crystal growth process is completed. The crystal is gradually cooled while being held in the afterheater.

5 [0018] [R₃Ga₅O₁₂, La₃R₂Ga₃O₁₂ or Gd₃R₂Ga₃O₁₂ Garnet]

A single crystal growth process for R₃Ga₅O₁₂, La₃R₂Ga₃O₁₂ or Gd₃R₂Ga₃O₁₂ is performed as follows.

Lanthanum oxide (La₂O₃), yttrium oxide (Y₂O₃), gadolinium oxide (Gd₂O₃), ytterbium oxide (Yb₂O₃), lutetium oxide (Lu₂O₃) and gallium oxide (b-Ga₂O₃) which have a purity of 5N

10 (99.999 %) are used as starting materials. These starting materials are checkweighed and mixed to obtain an intended composition, and then molded and sintered to form a raw material of a luminescent material. A single crystal is grown using the aforementioned modified micro pulling-down apparatus based on high-frequency induction heating, in the same manner as that of the above process, except that a Y₃Ga₅O₁₂ single crystal cleaved along the <111> direction is used

15 as a seed crystal.

[0019] [Li₆R(BO₃)₃ Borate]

A single crystal growth process for Li₆R(BO₃)₃ is performed as follows.

Boron oxide (B₂O₃), lithium carbonate (Li₂CO₃), yttrium oxide (Y₂O₃), gadolinium oxide (Gd₂O₃), ytterbium oxide (Yb₂O₃) and lutetium oxide (Lu₂O₃) which have a purity of 5N

20 (99.999 %) are used as starting materials. These starting materials are checkweighed and mixed to obtain an intended composition, and then molded and sintered to form a raw material of a luminescent material.

A single crystal is grown using the aforementioned modified micro pulling-down apparatus based on high-frequency induction heating, in the same manner as that of the above process, except

25 that a Li₆Y(BO₃)₃ single crystal cleaved along the <010> direction is used as a seed crystal.

[0020] The characteristics of the luminescent material of the present invention will be described below.

[Concentration Quenching]

In the emission based on the charge transfer state (CTS) between Yb (ytterbium) and O (oxygen), the concentration-dependent quenching is not severe (it comes to the front when the content of Yb is greater than 40 atomic%) as compared to that in the emission based on the $5d \rightarrow 4f$ transition of Ce (it comes to the front when the content of Ce is greater than 1 atomic%). Thus, the luminescent material of the present invention is advantageous in obtaining enhanced brightness using a luminescent element at a high concentration.

[0021] [Segregation]

Yb has an ionic radius approximately equal to those of typical rare-earth ions (Y, Gd, Lu) in a base crystal, and thereby the segregation coefficient of Yb is close to 1. Thus, if only the conditions of crystal growth are optimized, a single crystal can be prepared without any variation in composition in the crystal growth direction. This is significantly advantageous in application to a next-generation PET requiring a high degree of accuracy.

[0022] [Emission Wavelength]

As compared to the wavelength (~ 370 nm) of the emission based on the $5d \rightarrow 4f$ transition of Ce, the emission based on the charge transfer state of Yb in the luminescent material of the present invention has a long wavelength of ~ 540 nm. Thus, a semiconductor photodiode expected to provide a resolution about 30 to 40 times greater than that of a photomultiplier tube can be used as a photodetector around the wavelength (540 nm) of maximal sensitivity thereof.

That is, a combination of the luminescent material of the present invention and the semiconductor photodiode makes it possible to achieve a high-accuracy scintillation detector, and a PET using the scintillation detector can have an enhanced resolution capable of detecting a diseased region having a size of 0.1 mm.

[0023] [Luminescent-Decay Time]

While it has been considered that scintillator crystals generating emission based on the charge transfer state is unsuitable for PET requiring a high resolution because such an emission generally has a slow decay time, the emission based on the charge transfer state of Yb has a short decay time

due to temperature-dependent quenching at room temperature. A resulting light loss can be covered by the advantage of being capable of increasing the concentration of the luminescent element.

[0024] [Efficiency of Collision with Radiation]

5 The above single crystal serves as a scintillator crystal for converting radiation, such as γ -ray or X-ray, to visible light. It is reported that a scintillator with a higher density can absorb γ -ray or X-ray more efficiently. In this respect, Yb is a heavier element than Ce, and thereby advantageous in receiving a larger amount of radiation.

[0025] [EXAMPLE]

10 EXAMPLES 1 to 3

Table 1 shows the conditions of preparing Yb-containing mixed-crystal aluminum garnet single crystals in Examples 1 to 3 (Nos. 1 to 3).

[0026]

Table 1

No	Compound Composition	Starting Material	Melting Point, Atmosphere, Material of Crucible	Optimal Amount of Yb
1	{Y _{3-x} Yb _x }Al ₅ O ₁₂ x = 0.15 (5%) 0.30 (10%) 0.45 (15%) 0.60 (20%) 0.90 (30%)	Y ₂ O ₃ (4N) Yb ₂ O ₃ (4N) Al ₂ O ₃ (5N)	1980°C Ar flow Ir crucible	{Y _{2.55} Yb _{0.45} }Al ₅ O ₁₂ Yb = 15%
2	{Gd _{3-x} Yb _x }Al ₅ O ₁₂ x = 0.15 (5%) 0.30 (10%) 0.45 (15%) 0.60 (20%) 0.90 (30%)	Gd ₂ O ₃ (4N) Yb ₂ O ₃ (4N) Al ₂ O ₃ (5N)	2000°C Ar flow Ir crucible	{Gd _{2.55} Yb _{0.45} }Al ₅ O ₁₂ Yb = 15%
3	{Lu _{3-x} Yb _x }Al ₅ O ₁₂ x = 0.15 (5%) 0.30 (10%) 0.45 (15%) 0.60 (20%) 0.90 (30%)	Lu ₂ O ₃ (4N) Yb ₂ O ₃ (4N) Al ₂ O ₃ (5N)	2050°C Ar flow Ir crucible	{Lu _{2.55} Yb _{0.45} }Al ₅ O ₁₂ Yb = 15%

[0027] FIG. 2 is a photograph showing a single crystal of $\{\text{Lu}_{2.55}\text{Yb}_{0.45}\}\text{Al}_5\text{O}_{12}$ (Yb = 15 atomic%) in Example 3. FIG. 3 shows an X-ray-excited emission spectrum of the single crystal. FIG. 4 shows the decay time of the single crystal. The emission spectrum in FIG. 3 shows that the rightward peak is close to the wavelength range (530 nm) of maximal sensitivity of a semiconductor photodiode. This means that a semiconductor photodiode with sensitivity about 30 to 40 times greater than that of a conventional photomultiplier tube can be used as a photodetector in combination with the single crystal. This drastically enhanced sensitivity can contribute to improvement in resolution of PET.

[0028] EXAMPLES 4 to 8

Table 2 shows the conditions of preparing Yb-containing mixed-crystal gallium garnet single crystals in Examples 4 to 8 (Nos. 4 to 8).

[0029]

Table 2

No	Compound Composition	Starting Material	Melting Point, Atmosphere, Material of Crucible	Optimal Amount of Yb
4	$\{\text{Y}_{3-x}\text{Yb}_x\}\text{Ga}_5\text{O}_{12}$ x = 0.15 (5%) 0.30 (10%) 0.45 (15%) 0.60 (20%) 0.90 (30%)	Y_2O_3 (4N) Yb_2O_3 (4N) $\beta\text{-Ga}_2\text{O}_3$ (4N)	1750°C Ar + 0.2% O ₂ flow Ir crucible	$\{\text{Y}_{2.55}\text{Yb}_{0.45}\}\text{Ga}_5\text{O}_{12}$ Yb = 15%
5	$\{\text{Gd}_{3-x}\text{Yb}_x\}\text{Ga}_5\text{O}_{12}$ x = 0.15 (5%) 0.30 (10%) 0.45 (15%) 0.60 (20%) 0.90 (30%)	Gd_2O_3 (4N) Yb_2O_3 (4N) $\beta\text{-Ga}_2\text{O}_3$ (4N)	1780°C Ar + 0.2% O ₂ flow Ir crucible	$\{\text{Gd}_{2.55}\text{Yb}_{0.45}\}\text{Ga}_5\text{O}_{12}$ Yb = 15%
6	$\{\text{Lu}_{3-x}\text{Yb}_x\}\text{Ga}_5\text{O}_{12}$ x = 0.15 (5%) 0.30 (10%) 0.45 (15%) 0.60 (20%) 0.90 (30%)	Lu_2O_3 (4N) Yb_2O_3 (4N) $\beta\text{-Ga}_2\text{O}_3$ (4N)	1800°C Ar + 0.2% O ₂ flow Ir crucible	$\{\text{Lu}_{2.55}\text{Yb}_{0.45}\}\text{Ga}_5\text{O}_{12}$ Yb = 15%

7	$\{La_3\}[Lu_{2-x}Yb_x](Ga_3)O_{12}$ x = 0.10 (5%) 0.20 (10%) 0.30 (15%) 0.40 (20%) 0.60 (30%)	La ₂ O ₃ (4N) Lu ₂ O ₃ (4N) Yb ₂ O ₃ (4N) β-Ga ₂ O ₃ (4N)	1800°C Ar + 0.2% O ₂ flow Ir crucible	$\{La_3\}[Lu_{1.7}Yb_{0.3}](Ga_3)O_{12}$ Yb = 15%
8	$\{Gd_3\}[Lu_{2-x}Yb_x](Ga_3)O_{12}$ x = 0.10 (5%) 0.20 (10%) 0.30 (15%) 0.40 (20%) 0.60 (30%)	Gd ₂ O ₃ (4N) Lu ₂ O ₃ (4N) Yb ₂ O ₃ (4N) β-Ga ₂ O ₃ (4N)	1780°C Ar + 0.2% O ₂ flow Ir crucible	$\{Gd_3\}[Lu_{1.7}Yb_{0.3}](Ga_3)O_{12}$ Yb = 15%

[0030] EXAMPLES 9 to 11

Table 3 shows the conditions of preparing Yb-containing mixed-crystal borate single crystals in Examples 9 to 11 (Nos. 9 to 11).

5 [0031]

Table 3

No	Compound Composition	Starting Material	Melting Point, Atmosphere, Material of Crucible	Optimal Amount of Yb
9	$Li_6(Y_{1-x}Yb_x)(BO_3)_3$ x = 0.05 (5%) 0.10 (10%) 0.15 (15%) 0.20 (20%) 0.30 (30%)	Li ₂ CO ₃ (4N) Y ₂ O ₃ (4N) Yb ₂ O ₃ (4N) B ₂ O ₃ (4N)	1780°C 80% N ₂ + 20% O ₂ flow Pt crucible	$Li_6(Y_{0.85}Yb_{0.15})(BO_3)_3$ Yb = 15%
10	$Li_6(Gd_{1-x}Yb_x)(BO_3)_3$ x = 0.05 (5%) 0.10 (10%) 0.15 (15%) 0.20 (20%) 0.30 (30%)	Li ₂ CO ₃ (4N) Gd ₂ O ₃ (4N) Yb ₂ O ₃ (4N) B ₂ O ₃ (4N)	1780°C 80% N ₂ + 20% O ₂ flow Pt crucible	$Li_6(Gd_{0.85}Yb_{0.15})(BO_3)_3$ Yb = 15%
11	$Li_6(Lu_{1-x}Yb_x)(BO_3)_3$ x = 0.05 (5%) 0.10 (10%) 0.15 (15%) 0.20 (20%) 0.30 (30%)	Li ₂ CO ₃ (4N) Lu ₂ O ₃ (4N) Yb ₂ O ₃ (4N) B ₂ O ₃ (4N)	1780°C 80% N ₂ + 20% O ₂ flow Pt crucible	$Li_6(Lu_{0.85}Yb_{0.15})(BO_3)_3$ Yb = 15%

[0032] FIG. 5 is a photograph showing a single crystal of $\text{Li}_6(\text{Y}_{1-x}\text{Yb}_x)(\text{BO}_3)_3$ ($\text{Yb} = 15$ atomic%) in Example 9. FIG. 6 shows an X-ray-excited emission spectrum of the single crystal. As seen in FIG. 5, the decay time is about 30 ns. This decay time is approximately equal to that (10 to 80 ns) of Ce-doped GSO (Gadolinium Ortho-Silicate) or LSO (Lutetium Ortho-Silicate). This short
5 decay time can also contribute to improvement in resolution of PET.

BRIEF DESCRIPTION OF DRAWINGS

FIG. 1 is a schematic diagram showing the coordination of oxygen to positive ions, in a compound represented by the following composition formula: $\{\text{C}_3\}[\text{a}_2](\text{d}_3)\text{O}_{12}$.

10 FIG. 2 is an optical photograph showing the transparency of a single crystal of $(\text{Lu}_{2.55}\text{Yb}_{0.45})\text{Al}_5\text{O}_{12}$ ($\text{Yb} = 15$ atomic%) in Example 3.

FIG. 3 is a graph showing an X-ray-excited emission spectrum of the single crystal of $(\text{Lu}_{2.55}\text{Yb}_{0.45})\text{Al}_5\text{O}_{12}$ ($\text{Yb} = 15$ atomic%) in Example 3.

15 FIG. 4 is a graph showing the decay time of the single crystal of $(\text{Lu}_{2.55}\text{Yb}_{0.45})\text{Al}_5\text{O}_{12}$ ($\text{Yb} = 15$ atomic%) in Example 3,

FIG. 5 is an optical photograph showing the transparency of a single crystal of $\text{Li}_6(\text{Y}_{1-x}\text{Yb}_x)(\text{BO}_3)_3$ ($\text{Yb} = 15$ atomic%) in Example 9.

FIG. 6 is a graph showing an X-ray-excited emission spectrum of the single crystal of $\text{Li}_6(\text{Y}_{1-x}\text{Yb}_x)(\text{BO}_3)_3$ ($\text{Yb} = 15$ atomic%) in Example 9.